

# Design and Fabrication of Materials with Desired Deformation Behavior

Bernd Bickel<sup>1</sup> Moritz Bächer<sup>2</sup> Miguel A. Otaduy<sup>3</sup> Hyunho Richard Lee<sup>2</sup>  
Hanspeter Pfister<sup>2</sup> Markus Gross<sup>1,4</sup> Wojciech Matusik<sup>1</sup>  
<sup>1</sup>Disney Research Zurich <sup>2</sup>Harvard University <sup>3</sup>URJC Madrid <sup>4</sup>ETH Zurich



**Figure 1:** Two examples of real and replicated objects. Thanks to our data-driven process, we are able to measure, simulate, and obtain material combinations of non-linear base materials that match a desired deformation behavior. We can then print those objects with multi-material 3D printers using two materials (blue and black) with varying internal microstructure.

## Abstract

This paper introduces a data-driven process for designing and fabricating materials with desired deformation behavior. Our process starts with measuring deformation properties of base materials. For each base material we acquire a set of example deformations, and we represent the material as a non-linear stress-strain relationship in a finite-element model. We have validated our material measurement process by comparing simulations of arbitrary stacks of base materials with measured deformations of fabricated material stacks. After material measurement, our process continues with designing stacked layers of base materials. We introduce an optimization process that finds the best combination of stacked layers that meets a user's criteria specified by example deformations. Our algorithm employs a number of strategies to prune poor solutions from the combinatorial search space. We demonstrate the complete process by designing and fabricating objects with complex heterogeneous materials using modern multi-material 3D printers.

**Keywords:** fabrication, deformable objects, goal-based material design

## 1 Introduction

Elastic deformations are present in many objects in our everyday life, such as garments and shoes, furniture, plants, or even our own tissue. When we want to design elastic deformable objects, either in computer animation or in the real world, we are faced with the

problem of determining material descriptions and parameters such that the objects behave in a desired way.

Deformation effects can be modeled at very diverse scales, ranging from molecular interactions to globally-supported response functions, and through continuum elasticity laws or lumped-parameter models [Zohdi and Wriggers 2004]. As recently demonstrated by work in numerical coarsening and homogenization, the behavior of materials with microscale inhomogeneities can be approximated by mesoscale homogeneous materials [Kharevych et al. 2009]. We are interested in a process that can be regarded as the inverse of homogenization, i.e., we want to achieve a desired large-scale behavior through appropriate combination of materials at a smaller scale. Our approach fits in the category of goal-based design of deformable models, and it addresses the challenges of anisotropic, inhomogeneous, and non-linear behavior. Furthermore, it goes a step beyond computer animation and enables the physical fabrication of deformable materials with a desired behavior.

Our work is motivated by the recent development of multi-material 3D printers such as the OBJET Connex series [OBJET]. These printers are capable of manufacturing a variety of soft and hard materials with complex internal structures, making it suddenly possible to fabricate complex 3D objects with aggregate materials quickly, inexpensively, and accurately. This development offers new opportunities and challenges for physics-based animation research. Computer graphics has already contributed systems for designing and fabricating clothes [Okabe et al. 1992], plush objects [Mori and Igarashi 2007], paper craft objects [Mitani and Suzuki 2004], or surface microgeometry [Weyrich et al. 2009]. But there is a lack of tools and algorithms for designing, editing, and fabricating user-specified deformable objects.

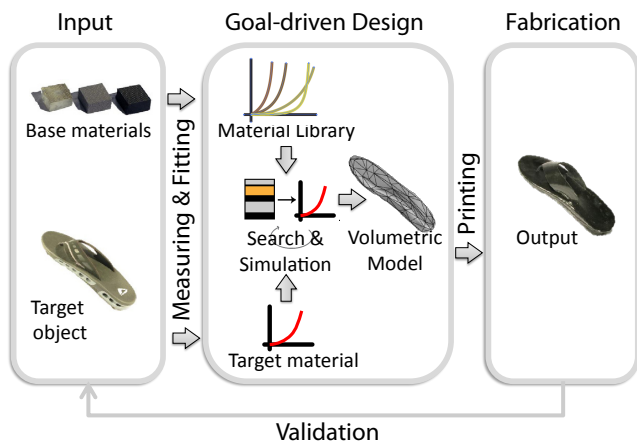
In this paper we make three main contributions: (i) An algorithm and representation for coarsening deformable models with microscale inhomogeneous behavior. Our solution uses a data-driven approach and is capable of capturing non-linear behavior while maintaining runtime efficiency. Conceptually, this can be seen as an extension of the linear numerical coarsening approach by [Kharevych et al. 2009] to non-linear material behavior. (ii) A goal-based material design approach that approximates a desired mesoscale deformation behavior by microscale materials through

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**Figure 2:** Designing deformable materials.

combinatorial optimization. This is an inverse modeling approach, inverting the concept of homogenization. (iii) A *complete* reproduction process for deformable materials, including acquisition, fitting, efficient simulation, goal-based design, and fabrication. The result of our design process serves as input to a 3D multi-material printer for the actual physical fabrication of deformable objects.

Our approach to measure, design, and fabricate materials with desired deformation behavior has a number of distinct steps, summarized in Fig. 2. We have collected a database of base materials, fabricated using a Connex 500 multi-material 3D printer, but also a variety of standard foams, gels, and rubbers purchased from the McMaster-Carr catalogue. These materials span a wide gamut of different deformations: from very soft to very hard and rigid (Section 7). We automatically measure deformations of these base materials subject to different forces using a robotic system (Section 6).

Next, we apply our coarsening algorithm to model the base materials using a data-driven non-linear stress-strain relationship in a Finite Element Method (FEM) (Section 3 and 4). This compact representation allows us to predict deformations of thicker or thinner versions of the base material samples. More importantly, we show that we can accurately predict deformations of arbitrary combinations of stacked base materials.

As the last step, we design composite materials that best match a desired deformation behavior using our combinatorial optimization algorithm (Section 5). In order to simplify the material design process, we introduce a goal-based optimization approach. The user specifies a material by providing example deformations and their corresponding forces, and our algorithm automatically computes the best-matching composite material. Because the configuration space is combinatorial and exponentially large, we use an efficient search strategy that prunes away states that yield poor matches to the desired material specifications.

We validate the simulation and material model by fabricating a number of different composite materials, measuring their deformations subject to a variety of different forces and comparing these measurements to the results of the simulation. The results are described in Section 7.

## 2 Related Work

In the context of computer graphics, our work is closest to the simulation of soft tissue using finite element models, and to data-driven modeling of deformable materials. For a recent survey of deformation models in computer graphics, please refer to [Nealen et al. 2006].

The most popular approach nowadays for accurately modeling deformable materials in computer graphics is probably to use continuum elasticity laws together with finite element modeling. This approach is capable of modeling a large range of materials, including those with nonlinear and heterogeneous deformation behavior, such as the base materials used in our work. Typically, one must select a constitutive material model [Ogden 1997] that is capable of covering the range of behaviors of the material, and then, given a certain object and constitutive model, the material parameters are tuned in order to fit empirical data. This approach was introduced to bio-mechanical modeling in computer graphics by Terzopoulos et al. [1987], and it has been later applied to body parts such as the face [Koch et al. 1996; Magnenat-Thalmann et al. 2002; Terzopoulos and Waters 1993; Sifakis et al. 2005], the hand [Sueda et al. 2008], the neck [Lee and Terzopoulos 2006], the torso [Zordan et al. 2004; Teran et al. 2005; DiLorenzo et al. 2008] or the complete upper body [Lee et al. 2009]; to the simulation of fracture effects [O’Brien and Hodgins 1999]; or even interactive simulation after a model reduction step [Barbič and James 2005]. In order to achieve high realism, continuum elasticity approaches rely on complex processes involving accurate modeling of the geometry and fine tuning of parameters.

Several researchers have designed methods to automatically identify the parameters of constitutive models from measurements of real objects. These measurement-based modeling approaches cover the estimation of parameters such as Young modulus [Schnur and Zabarar 1992], both Young modulus and Poisson ratio together [Becker and Teschner 2007], non-linearly interpolated Young modulus and Poisson ratio [Bickel et al. 2009], plasticity parameters [Kajberg and Lindkvist 2004], or non-linear viscoelasticity parameters [Kauer et al. 2002]. In this line of work, our data-driven technique for modeling base materials builds on the approach of Bickel et al., but with some notable differences. First, by restricting the types of materials to homogeneous ones, our model requires far fewer degrees of freedom. Homogeneity of the base materials is not a limitation in our case, since we achieve inhomogeneity in the final output materials by combining various homogeneous materials. Second, we increase the robustness of the fitting process by fitting one single non-linear model to all input examples simultaneously.

Other measurement-based modeling approaches fit directly globally-supported functions as material description instead of estimating local parameters. Pai et al. [2001] introduced a system for capturing in a unified framework an object’s shape, its elasticity, and roughness features. They used a Green’s functions matrix representation [James and Pai 1999] in order to describe the deformation model. Later, others have extended the work of Pai et al. to increase robustness and handle viscoelastic effects [Lang et al. 2002; Schoner et al. 2004].

Recent work in computer graphics aims at modeling high-resolution heterogeneities even when the resolution of the discretization is considerably coarser [Kharevych et al. 2009; Nesme et al. 2009]. This process, known as *homogenization*, tries to find parameter values of a constitutive model sampled at low resolution such that the behavior of the object best matches the heterogeneous material. The first step of our design and fabrication process can be considered as a variant of homogenization, where the fine-scale inhomogeneous material is an actual physical one. In the second step, however, we take the opposite approach to homogenization, generating a heterogeneous object that fits coarse force-deformation data from small-scale materials with known behavior.

A general introduction to optimizing spatial material distributions can be found in [Bendsoe and Sigmund 2003]. Digital materials, composed of a set of discrete voxels, can exhibit widely varying

material properties [Hiller and Lipson 2009]. As the design space increases exponentially with the number of possible combinations, evolutionary algorithms are a popular non-linear optimization strategy [Kicinger et al. 2005]. In contrast, we apply a branch-and-bound search strategy in combination with clustering.

### 3 Non-Linear Material Model

All our base materials exhibit a non-linear hyper-elastic stress-strain behaviour, as demonstrated by the measured force-displacement curves in Fig. 5. Most of the base materials consist of complex structures on a microscale level which influences the deformation behavior significantly (see Fig. 6). We use a data-driven approach and capture such non-linear behavior by a non-linear interpolation of locally linear material properties obtained from example deformations of the base material at a mesoscale. Conceptually, this can be seen as a data-driven coarsening approach, representing the deformation behavior induced by microstructures on a mesoscale level.

To achieve this, we employ the concept of sampling the stress-strain function of the material at a small set of strain values, and represent each stress-strain point by a linear co-rotational model [Bickel et al. 2009]. In the deformed state, at a given point in the object and with an arbitrary strain value, we represent the material locally by a non-linear interpolation of the sampled linear models.

Our linear FEM relies on Cauchy's linear strain tensor. Given a displacement field  $\mathbf{u}$ , it is defined as  $\varepsilon(\mathbf{u}) = \frac{1}{2}(\nabla\mathbf{u} + (\nabla\mathbf{u})^T)$ . We obtain invariance of the strain under rotations by extracting the rotational part of the deformation gradient through polar decomposition, and then warping the stiffness matrix [Müller and Gross 2004].

Due to the symmetry of the strain and stress tensors, we can represent both as 6-vectors. Given the strain tensor, we construct its equivalent 6-vector as  $\varepsilon = (\varepsilon_{xx} \ \varepsilon_{yy} \ \varepsilon_{zz} \ \varepsilon_{xy} \ \varepsilon_{xz} \ \varepsilon_{yz})^T$ , and similarly for the stress. Then, a linear material can be represented by a  $6 \times 6$  matrix  $\mathbf{E}$  that relates the stress and strain vectors,  $\sigma(\mathbf{u}) = \mathbf{E}\varepsilon(\mathbf{u})$ .

The key for achieving the non-linear behavior of the base materials is to define the stress-strain relationship,  $\mathbf{E}$ , as a function of the local strain  $\varepsilon(\mathbf{u})$ . The matrix  $\mathbf{E}$  can be typically parameterized by a smaller set of parameters  $\mathbf{p}$ . We define such parameters as a non-linear function of strain, therefore, the matrix  $\mathbf{E}$  can be represented as a non-linear function  $\mathbf{E}(\mathbf{p}(\mathbf{u}))$ . Although this approach holds for general anisotropic behaviour, we describe subsequently the parameters  $\mathbf{p}$  for two types of materials that are most relevant in practice: isotropic and transversely isotropic ones. The transversely isotropic material will be further discussed in Section 6.

**Isotropic Materials** For homogeneous linear isotropic materials, the matrix  $\mathbf{E}$  can be represented by the two Lamé parameters  $\lambda$  and  $\mu$ . In other words, the parameter vector is  $\mathbf{p} = (\lambda, \mu)$ . Using Lamé's parameters, the stress and strain tensors are related as

$$\sigma(\mathbf{u}) = 2\mu\varepsilon(\mathbf{u}) + \lambda \text{tr}(\varepsilon(\mathbf{u}))\mathbf{I}, \quad (1)$$

from which the matrix  $\mathbf{E}$  can be derived [Bathe 1995].

In homogeneous linear isotropic materials, the strain can be well captured by the three invariants of the symmetric strain tensor  $I_1(\varepsilon)$ ,  $I_2(\varepsilon)$ ,  $I_3(\varepsilon)$ . These invariants do not change under rotation of the coordinate system. Using the invariants to represent the strain, our non-linear material model can be considered as a 2-valued function in a 3-dimensional domain,  $\mathbf{p}(I_1, I_2, I_3) : \mathbb{R}^3 \rightarrow \mathbb{R}^2$ .

**Non-Linear Interpolation of Material Properties** Given a base material, we describe its non-linear stress-strain relationship through a small set of  $P$  parameter vectors,  $\{p_i\}$ , corresponding to different strain values,  $\{\varepsilon_i\}$ . Then, using the (parameter, strain) pairs as centers of Radial Basis Functions (RBF), we define the complete material behavior through RBF interpolation [Bickel et al. 2009]:

$$\mathbf{p}(\varepsilon) = \sum_{i=1}^P \mathbf{w}_i \cdot \varphi(\|\varepsilon - \varepsilon_i\|). \quad (2)$$

Since our base materials are homogeneous, a single set of parameter vectors is sufficient to describe the behavior of an arbitrary object consisting of a single base material. This reduces the number of parameters of a base material to  $|\mathbf{p}| \cdot P$ , where  $|\mathbf{p}|$  is the cardinality of the parameter vector (2 for isotropic materials, and 5 for transversely isotropic ones). In our examples, the number of RBF centers is typically between  $P = 6$  for the isotropic foams and  $P = 12$  for printed materials with complex internal microstructure. Computing the RBF interpolation based on the local strain in a spatially varying manner allows us to simulate different non-linear behavior in different regions of an object.

In order to simulate the behavior of composite objects made of base materials, we follow the quasi-static FEM approach of Bickel et al. [2009]. Given a simulation state, we compute the strain of all elements and perform a per-element computation of the parameter vector according to Eq. (2). We then recompute the per-element stiffness matrices, and perform a new step of the FEM simulation.

### 4 Fitting Base Materials

We fit the properties of base materials such that simulated deformations match best a set of input examples. In the fitting process, we need to compute the RBF centers  $\{\varepsilon_i\}$  (i.e., strain values used as data points), and their corresponding weights  $\mathbf{w}_i$  (See Eq. (2)).

Let us first assume that the  $P$  RBF centers are known. Given a set of example deformations with measured displacements  $\{\bar{\mathbf{x}}_i\}$  and corresponding forces  $\bar{\mathbf{F}}_i$ , we compute the RBF weights  $\mathbf{w}$  by minimizing the error in the displacements, as:

$$\hat{\mathbf{w}} = \arg \min_{\mathbf{w}} \left\{ \sum_{i=1}^n \|\mathbf{x}_i(\mathbf{p}, \bar{\mathbf{F}}_i) - \bar{\mathbf{x}}_i\|^2 \right\}. \quad (3)$$

In order to define the RBF centers, we first fit a homogeneous linear material to obtain a constant set of material parameters. Using these parameters, we run FEM simulations for all measurements, and record strain values. We select the RBF centers by sampling the strain space with  $P$  points that cover the range of measured values. Using these RBF centers we can fit the material parameters, but we run several iterations to obtain a better coverage of the strain space.

There are two main differences between our material fitting strategy and the one proposed by Bickel et al. [2009]. First, since the base materials are homogeneous, the RBF weights are not spatially-varying, and the size of the problem reduces to  $|\mathbf{p}| \cdot P$ . Second, the objective function is defined by grouping the measured displacements of all example deformations at once. These two differences lead to improved robustness and fitting accuracy.

We use Levenberg-Marquardt optimization and compute the Jacobians as defined by [Bickel et al. 2009]. An unconstrained optimization problem may lead to material parameters that are not physically correct. In the case of isotropic materials, it is easy to bound Lamé's parameters by computing the Young modulus and Poisson

ratio and projecting these to physically valid values. In the case of transversely isotropic materials, we make sure that the stiffness matrix is positive definite using the technique of [Rebonato and Jäckel 1999]. Otherwise, the FEM simulation could become unstable.

The measured forces  $\bar{\mathbf{F}}$  are normal to the surface. However, the contact area below the force probe also undergoes small tangential forces during the measurement process, and we found that these missing forces produce small fitting errors. We increase the fitting quality by computing the missing tangential forces that would produce a perfect match at the probed surface points, and then reintroduce the tangential forces as known forces in the optimization problem.

Section 7 evaluates the quality of the fit and reports error values by comparing the simulation of fitted base materials to measured example deformations.

## 5 Goal-Based Material Design

Our main contribution is a goal-based material design approach that approximates a desired mesoscale deformation behavior by microscale materials through combinatorial optimization. We now describe the optimization algorithm to obtain composite structures made of a set of base materials. Our algorithm receives as input a description of the object surface, examples of desired force-displacement pairs, and a set of base materials with known deformation properties expressed in our non-linear material model.

### 5.1 Formulation of the Problem

We formulate the design process as an optimization problem where we need to choose the distribution of  $M$  possible base materials inside the fabricated object such that it matches the input force-displacement examples.

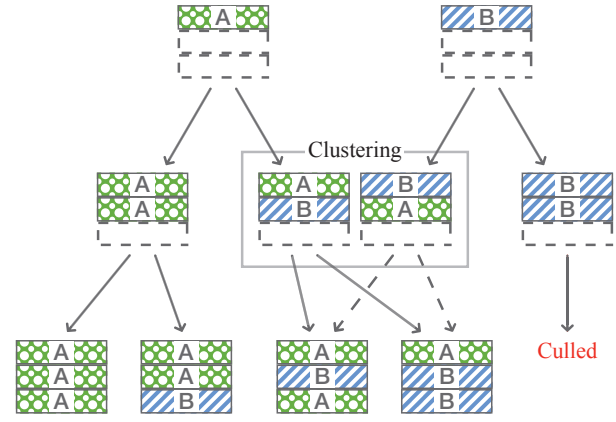
We discretize the problem by dividing the desired object shape in a set of  $N$  regular *cells*, each made of a uniform base material. The desired inhomogeneity and possible anisotropy of the final object are achieved by the appropriate distribution of base materials. For each cell, one may choose a single material from  $M$  possible base materials. We call a certain choice of base materials and their distribution a *design*. We denote each design as a vector  $\mathbf{m} = (m_1, m_2, \dots, m_N)$ , where  $m_i$  is an integer value that indicates the type of base material in the  $i^{\text{th}}$  cell out of the  $\{\bar{m}_j, 1 \leq j \leq M\}$  possible base materials.

In order to test each design, we assign its particular material choices to the cells of the object, simulate the object with the quasi-static FEM approach from Section 3 using the user-specified force profiles, and measure the error in surface displacements. The surface displacements of all input examples are grouped in one large vector  $\mathbf{x}$ . Given the goal displacements  $\bar{\mathbf{x}}$ , the *displacement error* of a design is simply  $\|\mathbf{x} - \bar{\mathbf{x}}\|$ . Finding the optimal design with minimal displacement error is an exponential problem, with  $M^N$  to-be-tested designs.

### 5.2 Branch-and-Bound with Clustering

The major problem when solving such a design optimization problem is the non-convexity of the design space and therefore the risk of ending up with a locally optimal solution if only the local neighborhood is observed [Lund and Stegmann 2005].

To solve this discrete optimization problem we therefore use a decision tree, such that at each level of the tree we span the options for one cell in the design. The root of the tree has  $M$  children, where



**Figure 3:** Branch-and-bound with clustering. The root of the tree shows the two materials A and B for the first out of three cells. Each level of the tree spans the possible options for the subsequent cells. Sub-optimal branches of the tree can be culled, and similar deformations can be clustered.

each child represents one of the material choices for the first cell, while the other  $N - 1$  remain undecided. Fig. 3 shows the decision tree for an object with three cells and two possible material choices.

Entire branches of the decision tree can be culled away using a branch-and-bound algorithm [Land and Doig 1960]. During tree traversal, we store the minimum error  $d_{min}$  for the designs tested so far. When a new node of the tree is visited, i.e., a new cell is refined, we use this minimum error to cull (if possible) the complete subtree rooted at the node.

Given the breadth of the tree, branch culling still leads to an intractable number of possible designs to be tested. However, often several designs produce similar deformation results. We cluster these nodes together to limit the breadth of the decision tree at every level, thus limiting the number of to-be-tested designs.

**Bound Estimation** Let us define as  $\{\mathbf{m}\}_a = (m_1, m_2, \dots, m_l, x \dots x)$  the designs rooted at a node  $a$  and located at level  $l$ . The first  $l$  cells are already determined along this branch, while the rest are still undecided (denoted by  $x$ ). We estimate a bound on the deformations produced by  $\{\mathbf{m}\}_a$  by considering the cases where the undecided cells are uniform. In other words, to estimate bounds we fill the undecided cells with each one of the base materials  $\bar{m}_j$ , compute the resulting deformation for all input examples  $\mathbf{x}_j$ , and then bound the result of the  $M$  cases as  $[\mathbf{x}_j]$ . We use axis-aligned bounding boxes in high dimensions as bounds, i.e., maximum and minimum values for each dimension of the resulting displacement vectors. We cull the branch rooted at node  $a$  if  $dist(\bar{\mathbf{x}}, [\mathbf{x}_i]) < d_{min}$ . When new designs are tested we update  $d_{min}$  appropriately.

The bound estimation approach is not conservative, due to material non-linearity and the existence of several non-monotonic functions along the simulation process. Recently, efficient methods for bounding displacements in linear FEM settings [Neumaier and Pownuk 2007] were presented, but practical bounds for non-linear settings are still an open research problem. However, the uniform blocks can be regarded as extreme behaviors (from very soft to very hard), therefore we expect that combinations of these materials will produce in-between deformations, in which case our bound estimation will not cull optimal designs.

**Clustering Strategy** We traverse the decision tree in a breadth-first manner, and hence a parent level with  $n$  nodes produces another level with  $n \cdot M$  nodes. Evaluating bounds on this new level requires the computation of  $n \cdot M^2$  designs. In order to limit the breadth of the tree, and thereby the total number of designs to be tested, we cluster nodes at every level before the splitting operation.

We cluster the  $n$  nodes at a level into  $K$  clusters using K-means clustering, using as distance  $dist(a, b)$  between two nodes the sum of squared example displacement differences evaluated for the pairwise uniform descendants. Formally, the distance metric is:

$$dist(a, b) = \sum_j^M \|\mathbf{x}(m_{1a}, \dots, m_{1a}, \bar{m}_j, \dots, \bar{m}_j) - \mathbf{x}(m_{1b}, \dots, m_{1b}, \bar{m}_j, \dots, \bar{m}_j)\|^2. \quad (4)$$

The cluster representative is the node that is closest to the centroid of the cluster. Every time we split a level we need to test only  $K M^2$  designs. Since the height of the tree is equal to the number of cells  $N$ , our clustering strategy limits the total number of design evaluations to roughly  $O(K M^2 N)$ . Note that the actual number of tested designs is actually smaller due to bound-based culling. In our implementation, we usually use  $K = 20$  clusters. This clustering approach comes at the cost of missing the global optimal solution.

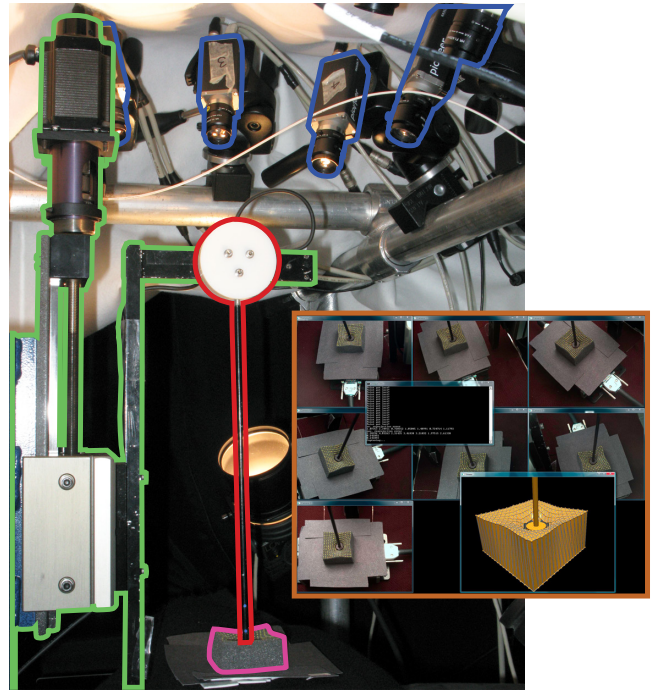
## 6 Measurement and Base Materials

**Measurement System** To acquire surface deformations of objects with a wide range of material properties we built an automatic measurement system that is able to acquire many different materials with varying geometry and surface properties. We use our system to probe base material samples, combinations of base materials for model validation, complex objects that we would like to reproduce, and their printed counterparts for validation purposes.

Our measurement setup (Fig. 4) consists of a four DOF robot arm (from MicroProto Systems), a six-axis force-torque sensor (Nano 25 from ATI), and a vision subsystem to track surface displacements. The resolution of the robot arm is 0.003 mm and its repeatability is 0.01 mm. The maximum range of the force sensor is 125 N with a resolution of 1/24 N. The vision subsystem consists of seven high-resolution Basler Pilot cameras running at a resolution of 1600 x 1200 pixels. We set up the calibrated ([Svoboda et al. 2005]) cameras on a half-circle above the robot arm to minimize occlusions and added diffuse lighting.

We paint marker dots on the surfaces of our objects in a regular grid with 3 mm spacing. The marker positions are extracted using a scale and affine invariant blob detector ([Mikolajczyk and Schmid 2004]) and tracked in all the frames. For each acquisition we use 30 to 200 deformation steps depending on the stiffness of the material. The maximal forces are in the range of 35 to 50 N. The tracked markers and corresponding forces are finally registered to a surface mesh.

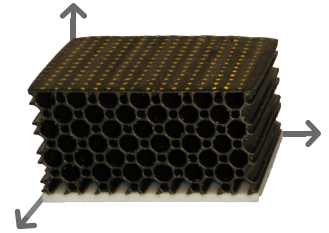
**Isotropic Base Materials** To print 3D deformable objects and a set of base materials we use the OBJET Connex 500 multi-material printer. In each run, the printer can use up to two different materials, e.g., Vero White (rigid) and Tango Black Plus (soft). The printer can also mix these two materials in predefined proportions producing isotropic materials of intermediate stiffness. We mainly use *Tango Black Plus* (TBP) and a mixed material called *digital material with shore 50* (DM50). In addition to these two isotropic base materials we measured eight isotropic materials from the McMaster-Carr online catalog, including rubbers and foams.



**Figure 4:** The automated system for measurement of material deformations consists of cameras (blue), a robot arm (green), and a force sensor attached to a stick (red). A sample material block is shown in pink and the inset shows a screen shot of our processing software.

Fig. 5 shows a plot of surface displacement as a function of applied force for a subset of measured materials.

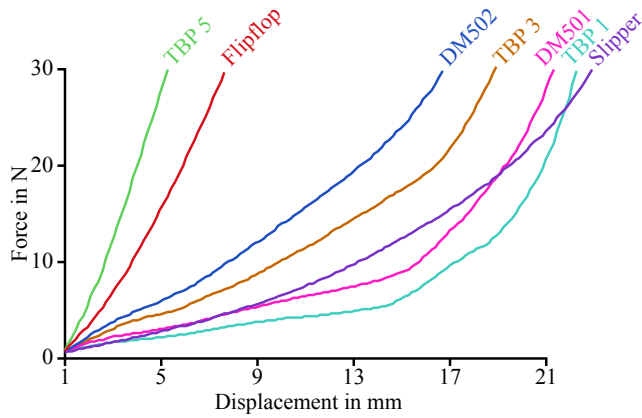
**Transversely Isotropic Base Materials** In order to model and fabricate materials with even larger deformation gamut (in particular, materials that are much softer) we introduce internal empty spaces into the printed objects. Unfortunately, the current printer only allows printing empty spaces that span the entire object along the z-axis. We use tubes of four different sizes and distributions (see example on the right). These objects are isotropic in the horizontal plane, perpendicular to the tube direction. The material can be regarded as *transversely isotropic*.



For such materials, the matrix  $\mathbf{E}$  can be represented as:

$$\mathbf{E} = \begin{bmatrix} E_{11} & E_{12} & E_{13} & 0 & 0 & 0 \\ E_{12} & E_{11} & E_{13} & 0 & 0 & 0 \\ E_{13} & E_{13} & E_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & E_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & E_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{(E_{11}-E_{12})}{2} \end{bmatrix}. \quad (5)$$

with five degrees-of-freedom,  $\{E_{11}, E_{12}, E_{13}, E_{33}, E_{44}\}$ . Our non-linear material model can then be considered as a five-valued function in a six-dimensional strain domain,  $\mathbf{p}(\boldsymbol{\varepsilon}) : \mathbb{R}^6 \rightarrow \mathbb{R}^5$ .



**Figure 5:** Force-displacement curves for a subset of the measured materials showing their nonlinear behavior.

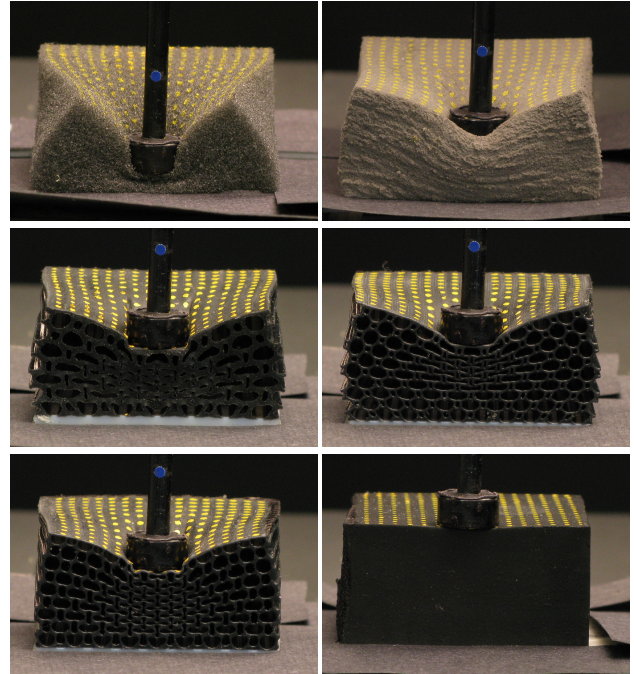
All base materials were printed as 4 cm (width)  $\times$  5 cm (length)  $\times$  2.5 cm (height) blocks. The deformations (side view) of some of these materials under 15 Newtons force are shown in Fig. 6. We will make the full set of base materials including all measurements and specifications public and plan to measure more materials in the future.

## 7 Validation and Results

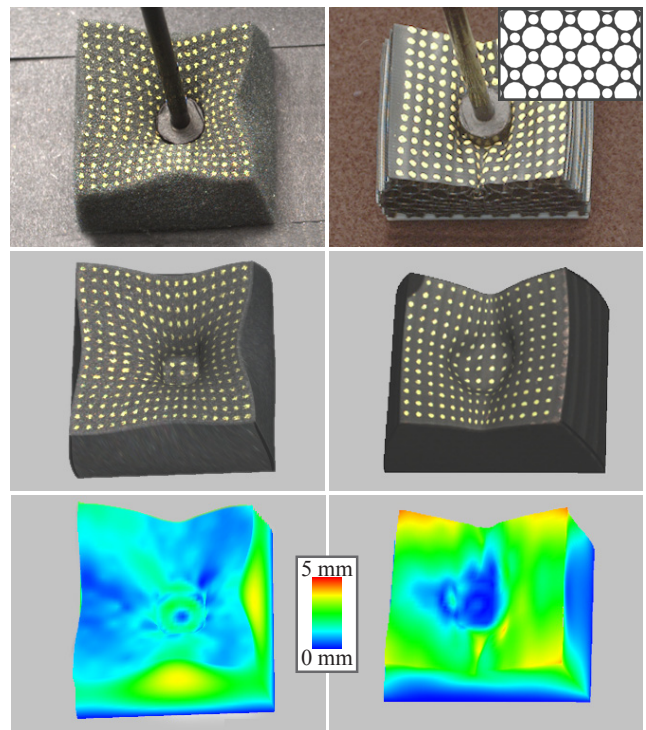
**Validation of the Fitting** Our material model represents elastic behavior of the base materials at the mesoscopic level very well. In Fig. 7 we compare images from our measurement system, the reconstructed deformed surface, and the corresponding simulation based on the FEM. We also show an error plot between the measured surface and the simulation. Note that the error is only evaluated at the surface marker positions and then interpolated for visualization purposes. Furthermore, the error evaluation is dependent on the accuracy of the measurement system which is in the range of  $< 1$  mm. Very small pitching effects at the microscale of the material cannot be tracked by our system and are therefore missing in the error visualization. Refer to the video for more results. For isotropic base materials we use six and for the transversal isotropic base materials 12 RBF centers, resulting in 12 and 60 parameters for each base material, respectively. The fit to the material model takes two hours on average but has to be performed only once. We also report the average, standard deviation, and maximum errors for the materials under varying applied loads in Table 1.

**Validation of the Stacking** Next, we show that we can accurately predict the behavior of composite materials made from arbitrary combinations of base materials. We ran a number of simulations for different composites and also fabricated those using the Connex 500 printer. Next, we measured the behavior of these composite materials using our system and compared them to their corresponding simulations. We report this validation for a few example deformations and materials in Fig. 8 and in the companion video. In the composite example shown in Fig. 8, we obtain average errors of 1.98 mm and 2.16 mm under loads of 10 N and 20 N.

**Validation of the Goal-based Design** Next, we validate our goal-based design process. As the first step we tested it on materials that we know we can reproduce. We picked a given combination of layers and their thicknesses. Then we simulated this composite material and used its deformations as the input to the



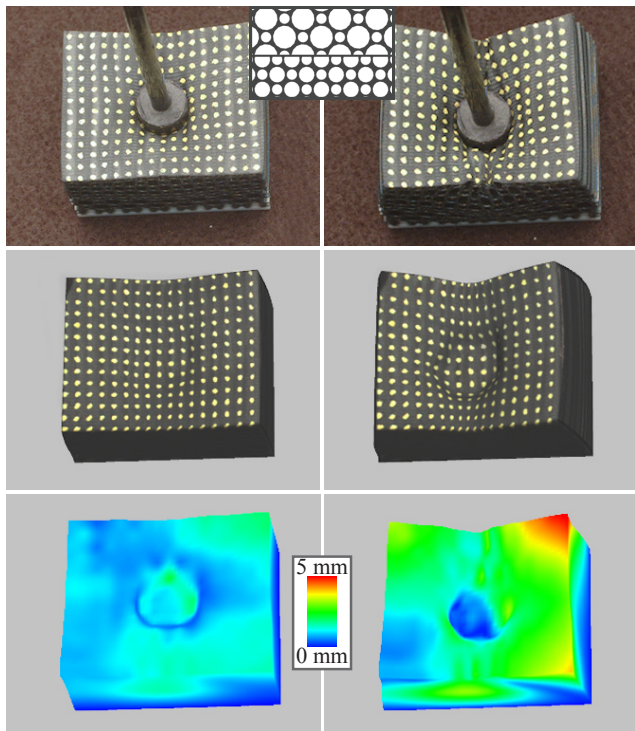
**Figure 6:** Pictures of base materials under 15 Newton force.



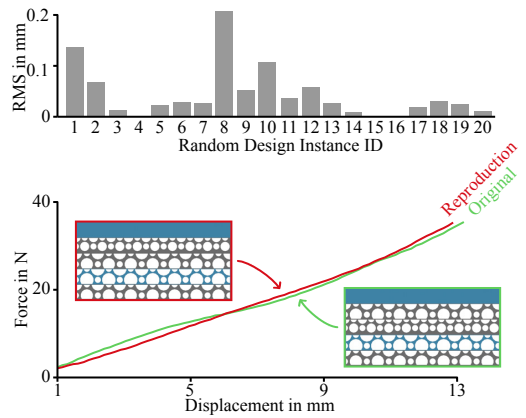
**Figure 7:** Side-by-side comparison of real and simulated materials. Deformation of an isotropic (left column) and transversely isotropic material (right column), comparing acquisition (top row) with the simulation (middle row) and the displacement error (bottom row).

Material	Force (N)	Displacement error (mm)		
		avg.	std.dev	max
Foam (very soft)	1N	0.84	0.45	2.55
	3N	1.72	1.00	6.74
	5N	2.04	0.88	5.70
Foam (medium)	5N	1.40	0.57	3.44
	25N	1.33	0.90	5.19
Foam (stiff)	10N	0.73	0.43	2.70
	20N	0.94	0.40	2.71
	30N	1.20	0.38	2.22
Printed TBP1 (soft)	5N	2.14	0.68	4.47
	20N	2.40	0.77	4.67
Printed DM502 (medium)	5N	0.69	0.26	1.44
	25N	1.31	0.51	3.00
Printed TBP5 (stiff)	10N	0.68	0.22	1.15
	20N	0.99	0.27	1.70
	30N	1.30	0.311	2.61

**Table 1:** Error evaluation of the model. We fitted parameters for various isotropic (soft/medium/hard foams) and transversely isotropic (printed materials with cylinder hole structures) materials and evaluated the surface displacement error under small, medium, and high force loads by comparing to measured deformations of material blocks (size isotropic 5x5x2.5cm, printed 5x4x2.5cm).



**Figure 8:** Validation of composite materials. We assigned the material properties obtained from two independent fits of base materials (DM501 and DM502) to a composite material consisting of two layers. We then printed the composite and compared the deformations of the real object (upper row) to the simulation (middle) under a load of 8 and 21 Newton. Lower row: Error visualization.



**Figure 9:** Validation of goal-based design algorithm. We randomly generate a set of material designs. We then simulate these designs and use their simulated deformations as the input to the goal-based design search algorithm. We then compare the obtained designs of the search algorithm with the known ground truth. The upper bar plot shows the RMS error. We also fabricated one of those randomly generated designs and its corresponding search output and compared their force-displacement curves.

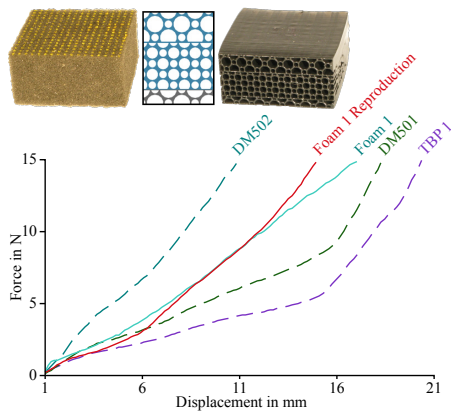
search algorithm. We report the result of this validation in Fig. 9. We tested this strategy on 20 different randomly chosen material designs (5 layers, each with 9 different material choices and 5 force-displacement pairs). Although our search is not guaranteed to find the global optimum, it always found a very close solution (average RMS error of 0.067 mm). The optimization time is usually below one hour. To carry this validation even further, we have fabricated these composites and remeasured their properties. We show these results in Fig. 9.

We also tried to approximate one of the foams with a combination of materials printed using the Connex 500. The obtained spatial combination and the error evaluation are shown in Fig. 10.

**Replicating objects** For the most complex results, we try to showcase the whole process by replicating objects with complex deformation behavior. In particular, we reproduced a beach flip-flop, a felt slipper, and a heterogeneous leather stool. We used a Cyberware scanner to scan the 3D geometry of each of the objects. Then, we acquired the deformation behavior of each object using our measurement system and fitted material parameters. For the leather stool, we segmented the volume into two areas, and approximated each of them as a homogeneous material. Next, we used the goal-based design process to find the best approximation of the material deformation properties using our base materials. For all results, we used between 5 and 10 force-displacement pairs. Finally, we printed replicas of these objects using the multi-material printer. As can be seen in Fig. 11 and 12 and the accompanying video the replicas show very similar behavior to the original objects. To further validate this approach we show force-displacement curves in Fig. 11 and 12 for the corresponding points on the replicas and the original objects.

## 8 Discussion

We have presented a complete approach for measuring, designing, and fabricating materials with desired deformation behavior. Our model is able to represent and simulate the non-linear elastic deformation behavior of objects with complex internal microstructure. In



**Figure 10:** Fabricated example of goal based design. As an input to our goal-based material design algorithm, we specified 5 example deformations of a foam (upper left). The desired deformation behavior is approximated by finer scale materials obtained through combinatorial optimization and then fabricated using a 3D printer (upper right). The lower curve shows the force-displacement relationship of used base materials, foam, and fabricated approximation.

order to ensure a good match between deformations of real materials and their simulated behavior, we use a data-driven measurement process to estimate non-linear stress-strain models for each material. Furthermore, we show that a goal-based material design approach can approximate a desired global deformation behavior by finer scale materials through combinatorial optimization. By closing the loop between measurements, simulation, goal-based material design, and printing, we validate the complete pipeline and show that close matches between simulated and real fabricated objects can be achieved. Our goal-based design is a significant step towards 3D hardcopying.

**Limitations and Future Work.** We believe that our system has many potential avenues for improvements and future work. We predict that this process will be a template for many future systems that expand the range of simulated and fabricated material properties (such as dynamic deformation properties or plasticity). More specifically, we plan to extend our model to dynamic and plastic deformation behavior and improve our measurement system such that it can acquire a wider range of deformation properties (e.g., material stretching and dynamic deformation measurements) or can guarantee and incorporate prior physical knowledge, such as volume preservation. Additionally, we plan to investigate strategies for optimally choosing the number of degrees of freedom (RBF centers) of our material model, striking a balance between accuracy and overfitting. Furthermore, we would like to examine material homogenization strategies to improve the speed of the forward (simulation) step for non-linear materials. This improvement along with more advanced search strategies could, in turn, speed up the inverse step such that we could design and fabricate extremely complex heterogeneous materials.

**Spatial combinations.** Currently, we only print layers of different materials. However, we believe our algorithm could be extended in a straight forward manner to arbitrary spatial combinations (e.g., voxels) of base materials. The decision tree could be directly applied to 2D or 3D problems, by having a one-to-one mapping of layers in 1D to voxels in 3D. Also, our pruning strategy (clustering and bounds) can be directly translated to the 3D case. Our search algorithm linearly scales with the number of layers or volume elements.

For current printers, the mechanical range of isotropic base materials without any holes or tube structure is limited. The OBJET Connex 500 printer can mix two different materials, and the material properties are restricted to the range between the two base materials. To significantly expand this range, we deliberately decided to create tube-structured materials. Due to current physical printer limitations, these void tube structures can only be printed along the z-axis of the printer, otherwise they would get filled with structure material, which is difficult to remove. Printing blocks or objects with isotropic hole structures (similar to Swiss cheese) is currently not possible. This comes at the cost of requiring a transversely isotropic material model.

Looking into the future, we predict that the next generation of 3D multi-material printers will be able to use many more base materials with a wider range of material properties and more complex internal structures. As the cost of these printers decreases and their capabilities increase, we believe that the goal of personalized design, modeling, simulation, and fabrication will become reality.

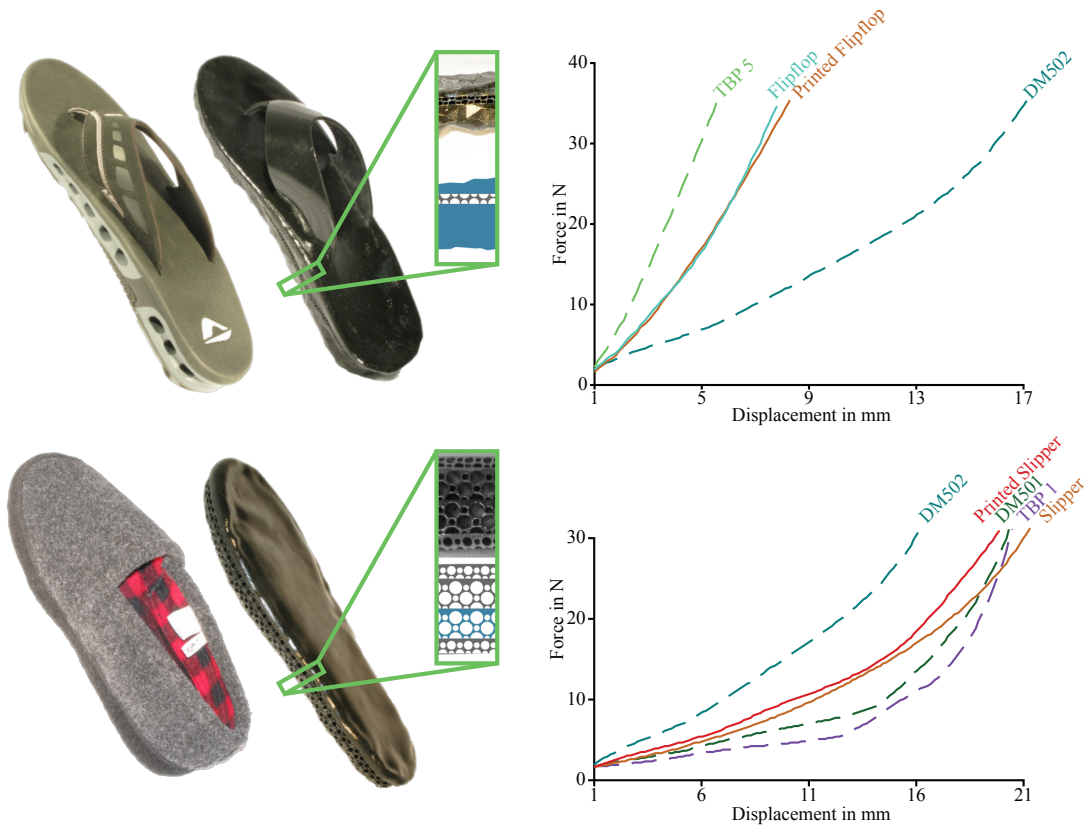
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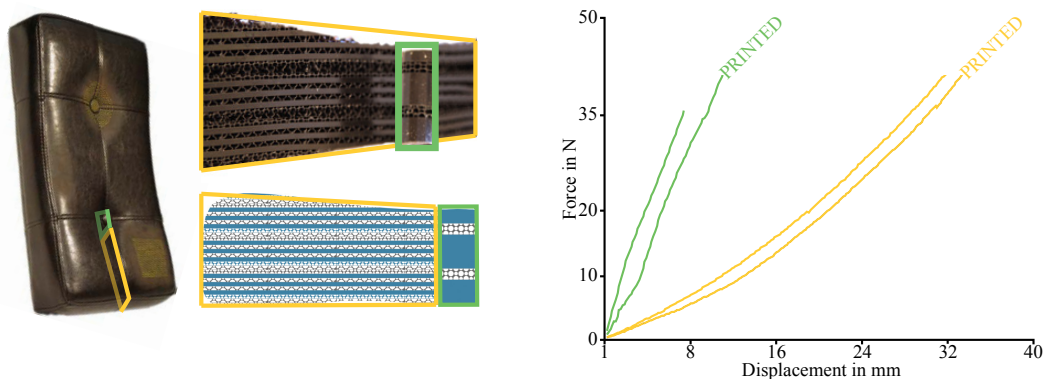
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**Figure 11:** Replicating a flip flop (upper row) and slipper (lower row). The left column shows the original images, the middle column the replicated flop flop and slipper with the spatial combination of base materials obtained by our goal-based optimization approach. Our replication matches the deformation behaviour of the original quite well, as shown in the force displacement plots (right column) for a corresponding point on the original and replica. The dotted curves characterize the base materials.



**Figure 12:** Replicating a leather stool. The left column shows the original object, the middle column a cross section of the replicated object and the spatial combination of base materials. We segmented the stool into two regions, a stiff region below the button (indicated in green) and the remaining softer region (indicated in orange). We validated the deformation behaviour by comparing the force displacement plots (right column) in the button region (orange) as well as in the softer region (green).

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